that the full-scale intensity of the phase I diffractometer scan was considerably lower than either of the other two samples (100 cps full-scale for phase I versus 2000 cps for the other samples). Since low-pressure-crystallized phase I exists only in the oriented state, the intensity of peaks obtained by the diffractometer techniques will be greatly affected. The (001) reflection in phase I which occurs at 2.55 Å ($2\theta = 35.16$) does not appear in the diffractometer scan due to this orientation. However, the peak positions should be correct for zero layer reflections of phase I, and, therefore, a comparison of peak positions in the three samples can be made. The (021) reflection does not appear in the scan of the high-pressure-crystallized sample in Fig. 2. This is undoubtably also due to orientation in this sample, since the (021) reflection (d = 3.34 Å; Table 1) appears in Debye-Scherrer photographs of samples crystallized at 5000 atm at high supercoolings. These scans point out the similarities between phases I and II and the high pressure crystallized PVF, samples in a manner that is much easier to interpret than if Debye-Scherrer photographs were compared. If, indeed, there is a mixture of phases I and II, then the fact that there are indications of two melting peaks for the high-pressure crystallized samples might be explained. It

TABLE 1

Qualitative Comparison of Intensities of X-Ray
Reflections Obtained from Debye-Scherrer Photographs^a

| Crystallization temperature (°C) | Degree of Supercooling (°C) | Intensity of peaks occurring near the following d spacings $(\mathring{A})^b$: | | | | |
|--|-----------------------------------|---|--------------|------|------|--------------|
| | | 5.00 | 4.81 | 4.45 | 4.27 | 3.34 |
| | Low-pressure- | | | | | |
| - | crystallized | M | VS | VS | Α | M |
| | PVF ₂ phase I | | | | | |
| 190 | 118 | W | VS | VS | Α | \mathbf{M} |
| 205 | 103 | VW | S | VS | W | W |
| 268 | 40 | A | \mathbf{M} | S | M | VVW |
| 282 | 26 | A | W | M | S | Α |
| 286 | 22 | A | VW | W | VS | Α |
| 290 | 18 | A | A | VVW | VS | Α |
| | Low-pressure- | | | | | |
| | crystallized phase I | A | A | Α | VS | A |

^aAll samples were crystallized at 5000 atm.

^bA = absent, VVW = very very weak, VW = very weak, W = weak, M = medium, S = strong, VS = very strong.

has already been shown earlier in the discussion that if the copolymers are high-pressure crystallized, the resulting samples have a higher melting point. By varying the degree of supercooling on samples crystallized at 5000 atm, the relative size of the melting peaks occurring at 160° and 187°C changes. This is illustrated in Fig. 3.

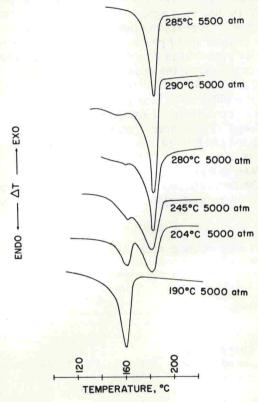


FIG. 3. DTA thermograms as a function of crystallization temperature for PVF, samples crystallized at 5000 atm.

A sample which was heated to 190° C and pressure-crystallized at 5000 atm had a diffractometer scan which was nearly identical to that of phase II (scan I in Fig. 2). The DTA melting point of 160° C for this high-pressure crystallized sample was also identical with that of the atmospherically crystallized PVF₂ phase II. The melting point thermogram for the sample which was heated to 280° C and pressure-crystallized at 5000 atm (scan 2 in Fig. 2) indicated melting endotherms at 160° and 187° C. The diffractometer scan for a